

Harmonic Resonance Enhanced Second-Harmonic Generation in the Monolayer WS<sub>2</sub>–Ag Nanocavity

Xiaobo Han, Kai Wang,\* Patrick D. Persaud, Xiangyuan Xing, Weiwei Liu, Hua Long, Fang Li, Bing Wang, Mahi R. Singh,\* and Peixiang Lu\*

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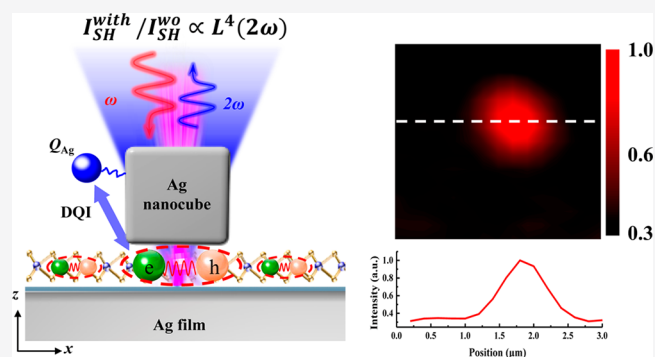
Supporting Information

**ABSTRACT:** The enhanced second-harmonic generation (SHG) from a monolayer WS<sub>2</sub> coupled to a plasmonic nanocavity is experimentally and theoretically investigated. The nanocavity is comprised of monodispersed Ag nanocubes separated from an Ag film by a spacer Al<sub>2</sub>O<sub>3</sub>, namely, the nanoparticle on mirror (NPoM) system. When the surface plasmon polariton resonance (SPPR) wavelength of NPoM nanocavity overlaps well with the SHG wavelength of the monolayer WS<sub>2</sub> (namely, harmonic resonance), a ~300-fold SHG enhancement is achieved in experiment. For theoretical understanding, the quantum mechanical density matrix method has been used to develop a theory for SHG. It is found that the SHG intensity of nanohybrid is proportional to the square of the local-field intensity in NPoM nanocavity at SHG wavelength, which is ascribed to the dipole–quadrupole interaction between dipole,  $P_{\text{SHG}}$ , in the monolayer WS<sub>2</sub> and quadrupole,  $Q_{\text{Ag}}$ , in Ag nanocavity. It is significantly different from that in metal nanoparticles under harmonic resonance, which is proportional to the local-field intensity. Therefore, it provides a novel mechanism for enhancing SHG signals from metal–semiconductor nanohybrids, which has potential applications in nonlinear devices and hybrid nonlinear metasurfaces.

**KEYWORDS:** second-harmonic generation, plasmonic nanocavity, monolayer WS<sub>2</sub>, Ag nanocube, dipole–quadrupole interaction

Second-harmonic generation (SHG) is an important second-order nonlinear optical response, which converts two photons of frequency  $\omega$  into another photon of double frequency  $2\omega$ .<sup>1</sup> Recently, SHG in semiconductor nanostructures, such as ZnS, GaAs nanowire, and monolayers MoS<sub>2</sub>,<sup>2–5</sup> has attracted much attention due to its potential applications in coherent nanosources,<sup>5,6</sup> imaging,<sup>7–9</sup> and nanoprobng.<sup>2,10–13</sup> However, the SHG process is generally weak at the subwavelength scale, since second-order nonlinear effects scale quadratically with the volume of the nanostructure.<sup>14</sup> Therefore, it is still faced with challenges to boost nonlinear conversion efficiency. Many works have focused on the enhanced nonlinear optical properties in the noble metal–semiconductor nanohybrids with a well designing.<sup>14–23</sup> Owing to a unique optical property called surface plasmon polaritons resonance (SPPR) in noble metals, the intensity of local-field can be efficiently concentrated which is ideal for enhancing nonlinear processes, such as SHG,<sup>18–22</sup> third-harmonic generation (THG),<sup>24</sup> and two-photon luminescence (TPL).<sup>25,26</sup>

In general, the SPPR wavelength in metal–semiconductor nanohybrids is usually designed to be resonant with the pumping wavelength, which is known as fundamental resonance. Since the fundamental local-field is boosted, the

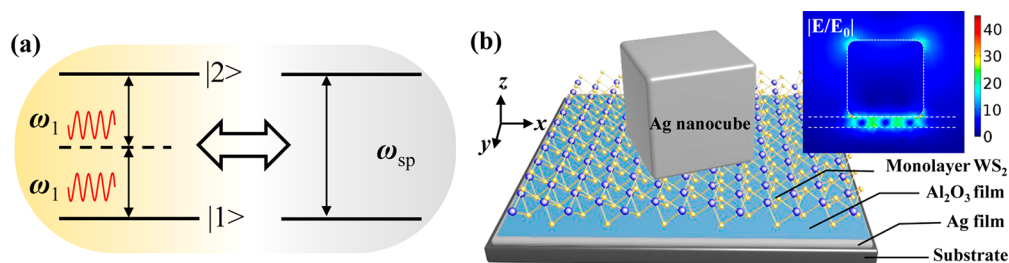


generated SHG signals can be efficiently enhanced by square of the fundamental local-field intensity.<sup>20,21</sup> However, the fundamental resonance can strongly enhance the absorption of the pumping laser, which may easily cause heat damage or unwanted signals, such as supercontinuum.<sup>27</sup> Therefore, a limited intensity of the pumping laser can be allowed. For improving SHG conversion efficiency, the SPPR wavelength is also tuned to match with the SHG wavelength shown in Figure 1a, which is called a harmonic resonance.<sup>28</sup> Since SHG intensity is in proportion to the local-field intensity at SHG wavelength, the SHG signal generally can only be enhanced by 1 order of magnitude.<sup>29,30</sup> As reported, a few works study the harmonic resonance enhanced SHG in pure metal nanoparticles (NPs).<sup>30–32</sup> To our knowledge, the harmonic resonance enhanced SHG in metal–semiconductor nanohybrids is still not addressed clearly.

On the other hand, two-dimensional transition-metal dichalcogenides (TMDs) have drawn a lot of interest in

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**Figure 1.** (a) An energy level schematic diagram of the hybrid metal–semiconductor system. The semiconductor material is made of two levels denoted as  $|1\rangle$  and  $|2\rangle$ , where  $|1\rangle$  is the ground state.  $\omega_1$  and  $\omega_{sp}$  denote the pumping laser frequency and SPPR frequency of metal NPs, respectively. When  $2\omega_1 = \omega_{sp}$ , there is a strong interaction between semiconductor materials and metal NPs. (b) A schematic diagram of the designed hybrid system. Plasmonic nanocavity is composed of single Ag nanocubes and an Ag film. A monolayer  $\text{WS}_2$  is placed in the nanocavity. The inset shows the calculated electric field of  $|E/E_0|$  in the  $xz$ -plane at resonances wavelength. White lines indicate the outline of the Ag nanocube with length of  $\sim 75$  nm.

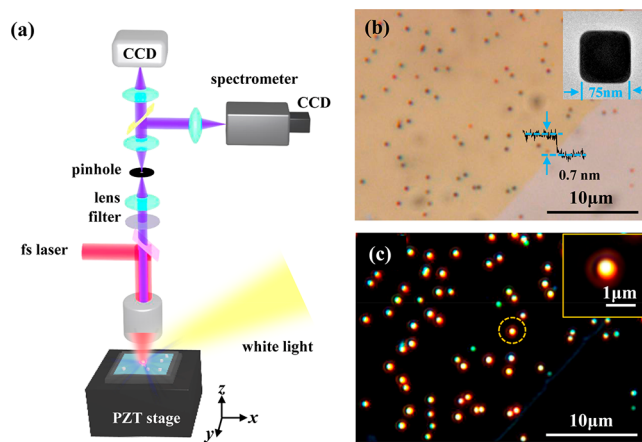
recent years due to the unique optical<sup>3,11,33</sup> and electrical properties.<sup>34</sup> Since the thickness of monolayer TMDs is less than 1 nm, many previously used nanohybrids are not applicable, such as core–shell and strip-covered structures.<sup>14,21,22</sup> Therefore, a highly confined nanocavity, namely, nanoparticle on mirror (NPOM), is introduced for monolayer TMDs. It is composed of monodispersed metal NPs separated from a metal film by a spacer, and a monolayer TMD is placed in the nanogap.<sup>35,36</sup> It has many advantages that cavity mode and resonance peaks can be easily tuned by NP size, shape, and the thickness of spacer layers.<sup>37–40</sup> It has been reported that the strong coupling<sup>41,42</sup> and the large PL enhancement in monolayer TMDs are observed with the nanocavity.<sup>43</sup>

In this work, Ag NPOM nanocavity are designed and coupled with monolayer  $\text{WS}_2$  to enhance the SHG signals. The nanocavity consists of Ag nanocubes over an Ag film, separated by an  $\text{Al}_2\text{O}_3$  layer ( $<10$  nm). By detuning the SPPR wavelength of nanocavity to the SHG wavelength at 410 nm, a large SHG enhancement factor (EF) of  $\sim 300$  is achieved experimentally. For understanding the enhancement mechanism, a theoretical model of semiconductor and nanocavity interaction is used. It demonstrates that the SHG intensity is proportional to the square of the local-field intensity of the Ag nanocavity at the SHG wavelength, which is caused by the dipole–quadrupole interaction between dipole,  $P_{\text{SHG}}$ , in the monolayer  $\text{WS}_2$  and quadrupole,  $Q_{\text{Ag}}$ , in the Ag nanocavity. It shows a significant difference with that in pure metal NPs under harmonic resonance, which is in proportion to the local-field intensity. Therefore, it reveals a clearly underlying mechanism for the SHG enhancement by harmonic resonance and provides further exploration for nonlinear enhancement of metal–semiconductor hybrid structures.

Figure 1b shows the schematic diagram of the monolayer  $\text{WS}_2$ –Ag NPOM nanocavity. It is composed of an Ag film, an  $\text{Al}_2\text{O}_3$  film, and a single Ag nanocube, where a monolayer  $\text{WS}_2$  is placed on the  $\text{Al}_2\text{O}_3$  film. Broadband tuning of the Ag nanocube plasmon resonance can be achieved due to strong electromagnetic coupling between the metal NPs and the film. An alternative approach to realize tuning the SPPR is to change the thickness of  $\text{Al}_2\text{O}_3$  spacer film. Since the thickness of the  $\text{Al}_2\text{O}_3$  film is usually quite small ( $<10$  nm), it leads to a strong local field in the NPOM nanocavity. The inset of Figure 1b shows the calculated local-field distribution in the NPOM nanocavity by COMSOL Multiphysics software. The plasmonic nanocavity was fabricated using a bottom-up method. First, a Ti/Ag thin film of thickness  $\sim 10/80$  nm was prepared on a silicon substrate via electron beam evaporation. Next,  $\sim 9$

nm  $\text{Al}_2\text{O}_3$  spacer film was deposited by atomic layer deposition (ALD) at the temperature of  $100^\circ\text{C}$ .  $\text{WS}_2$  flakes (SixCarbon Tech. Shenzhen) were grown on sapphires by chemical vapor deposition (CVD) method. Avoiding introducing dusts, we transferred  $\text{WS}_2$  flakes from the sapphire to the  $\text{Al}_2\text{O}_3$ –Ag film with the help of polydimethylsiloxane (PDMS).<sup>44</sup> Finally,  $10\ \mu\text{L}$  of diluted Ag nanocubes (nanoComposix,  $0.01\ \text{mg/mL}$ ) were directly dropped on the prepared substrate. After 1 min, the nanocubes solution was washed off with water, and the sample was dried with nitrogen gas. Ag nanocubes were monodispersed on the  $\text{Al}_2\text{O}_3$ –Ag film uniformly.

Figure 2a shows the schematic experimental setup for individual Ag nanocubes optical measurement. Figure 2b

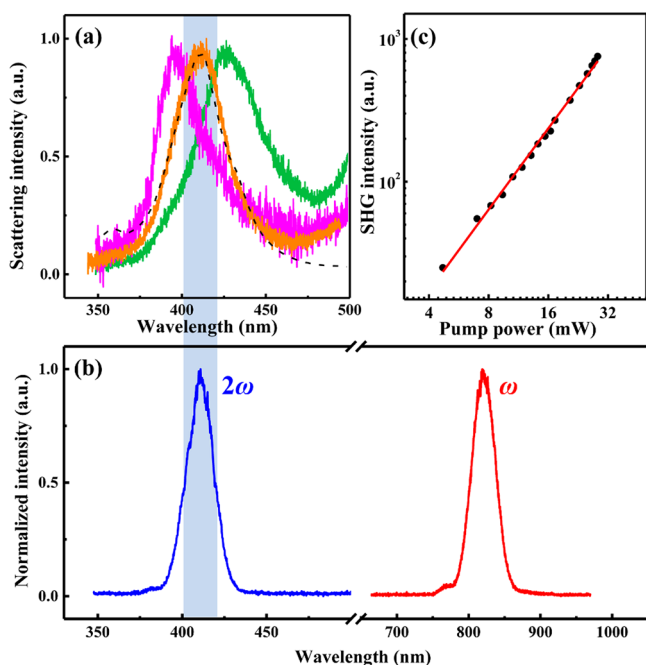


**Figure 2.** (a) Experimental setup for measuring dark-field scattering signals of individual NPs and SHG signals of the nanohybrids. An oblique incident white light was used for the scattering measurement. (b) Optical image of the nanohybrid, the yellow flakes represent the monolayer  $\text{WS}_2$ , and small dark spots represent individual Ag nanocubes. TEM image of an Ag nanocube and the thickness of a monolayer  $\text{WS}_2$  is shown in the insets. (c) Dark-field scattering image corresponding to (b). Bright spots represent individual Ag nanocubes corresponding to the dark spots in (b). The inset shows the NP image marked with a circle selected by a pinhole.

shows an optical microscope image of the nanohybrid: the yellow region is the area of the monolayer  $\text{WS}_2$ , while the small spots indicate individual Ag nanocubes. The inset in Figure 2b is the transmission electron microscopy (TEM, Tecnai G20) image of a single Ag nanocube with a length of  $\sim 75$  nm. Atomic force microscopy (AFM) measurement shows that the thickness of  $\text{WS}_2$  is  $\sim 0.7$  nm, indicating a monolayer  $\text{WS}_2$  is

used in experiment. Raman spectrum of WS<sub>2</sub> is shown in Figure S1, which is also confirmed for the monolayer nature. For dark-field scattering measurements, an oblique (about 45°) white light (Energetiq, EQ-99XFC) was focused by a lens of 5 cm and illuminated the sample. Scattering light was collected by an objective (Olympus, 50×, NA = 0.5, LMPlanFL N) and focused by a lens again. The focus plane falls on the image plane, where a pinhole (200 μm) was placed to choose individual NPs, as shown in the inset of Figure 2c. Then the scattering light through two lenses was sent to a CCD (Qimaging, QICAM B series) or a spectrometer (Princeton Instruments Acton 2500i with Pixis CCD camera). Figure 2c shows the dark-field scattering image of the same sample corresponding to Figure 2b. The size of Ag nanocubes is distributed from 65 to 95 nm (see Figure S2); therefore, different colors can be observed due to the varied scattering SPPR wavelength of Ag nanocubes with different sizes. The blue line in Figure 2c indicates the edge of the monolayer WS<sub>2</sub>. In order to avoid coupling among Ag nanocubes, a proper distribution of nanocubes is necessary.

Figure 3a shows the measured scattering spectra of single Ag nanocubes. It is worth noting that the single nanocavity has



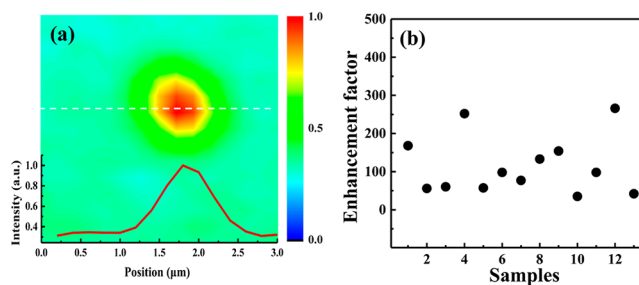
**Figure 3.** (a) Normalized dark-field scattering intensity of single nanocubes with different lengths (65–95 nm). A simulated scattering result is shown as a black dash-dot curve. (b) Measured laser spectrum (red) and its corresponding measured SHG spectrum (blue). (c) SHG intensity response as a function of the excitation (average) power on a log–log scale, which fits to a linear line with a slope about 2.

two SPPR wavelengths shown in Figure S3, so the scattering spots of Ag nanocubes in Figure 2c are not shown in blue colors. Due to size distribution of Ag nanocubes, the measured SPPR peaks are varied from ~400 to ~430 nm in Figure 3a. A black dash line indicates the simulated scattering spectrum, which is calculated by COMSOL Multiphysics software under a linear plane wave with a normal incidence. It is in good agreement with the experimental result (detailed parameters of the simulation are presented in the Supporting Information).

According to the definition of the quality factor,<sup>45</sup>  $Q = \omega_r / \Delta\omega$ , where  $\omega_r$  and  $\Delta\omega$  are the resonance frequency and full-width at half-maximum (fwhm), respectively, the  $Q$  factor is ~8 calculated from Figure 3a. Though it shows a relatively low  $Q$  factor of nanocavity, the nanoscale mode volume still results in a strong local-field.

The nonlinear optical properties of the monolayer WS<sub>2</sub> were measured by a home-built microscopy system, as schematically illustrated in Figure 2a. To measure the SHG signal of the monolayer WS<sub>2</sub>, a mode-locked Ti-sapphire femtosecond laser system (Virtara Coherent, ~820 nm, 20 fs and 80 MHz) was focused by a 100× objective (Olympus, NA = 0.8, LMPlanFL) to a spot diameter of ~1 μm. The radiated signals were collected by the same objective. A 720 nm short pass filter (Semrock, FF01–720/SP-25) was used to avoid the pumping laser. Another 400 nm band-pass filter was used before spectrometer to avoid TPL signals, when doing scanning tests. Same as the scattering optical path, the radiated signals through three lenses were sent to a CCD camera or a spectrometer. In Figure 3b, it indicates the measured optical signal (blue curve) is located at ~410 nm in the monolayer WS<sub>2</sub> on the Al<sub>2</sub>O<sub>3</sub>–Ag film, which is exactly twice the frequency of the pumping laser (red curve) at 820 nm. Furthermore, the signal intensity as a function of the pumping power is plotted in Figure 3c. The linear fit with a slope of about 2 indicates the quadratic dependence between the signal intensity and the pumping power. Thus, it is confirmed that the measured signal is attributed to SHG characteristics. Through a reflective neutral density filter (Thorlabs, NDC-50C-4M), the laser intensity in experiment is controlled below 10 kW cm<sup>-2</sup> to avoid photo damage of nanohybrids. Figure S4 shows the scattering spectra of nanohybrids before and after SHG measurement. Note that no SHG responses from Ag nanocubes or Ag films can be obtained under a relatively low pumping in experiment.

In order to investigate the SHG EF, the SHG scanning measurement is performed. A PZT stage (XMT, XE501-D) was used to move samples in  $xy$ -plane with a step of 200 nm. Since the average local-field intensity near Ag nanocubes is independent of the laser polarization (see Supporting Information for details), a linearly polarized pumping laser is used. Figure 4a shows the SHG intensity map (3 × 3 μm<sup>2</sup>) of the nanohybrid. It can be observed that the SHG intensity with the Ag nanocube is much stronger than that without nanocube (monolayer WS<sub>2</sub> on the Al<sub>2</sub>O<sub>3</sub>–Ag film). For a specific



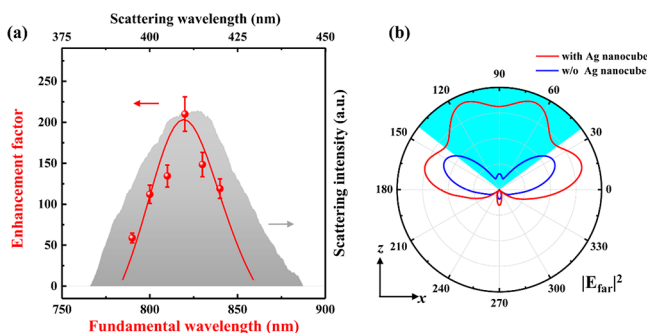
**Figure 4.** (a) Normalized SHG intensity map of a monolayer WS<sub>2</sub> inserted in a plasmonic nanocavity with a scanning step of 200 nm. The displayed scanning area is 3 μm × 3 μm. The inset curve shows extracted-line (white dash line) intensity marked in the map. (b) SHG EF with different Ag nanocubes under the fixed excitation laser with wavelength of 820 nm.

comparison, SHG intensity profile across the center of the Ag nanocube is shown in Figure 4a (white dashed line). By comparing the SHG intensity with and without the Ag nanocube, it indicates a 3-fold SHG enhancement. Considering the fact that the size of the Ag nanocube is much smaller than the diameter of the pumping focal spot ( $\sim 1 \mu\text{m}$ ), the averaged SHG EF across a single nanocavity is defined as

$$\text{EF} = \left( \frac{I_{\text{cavity}}}{I_0} - 1 \right) \frac{S_0}{S_{\text{cavity}}} \quad (1)$$

where  $I_{\text{cavity}}$  is the SHG intensity from the monolayer  $\text{WS}_2$  placed in Ag nanocavity and  $I_0$  is the SHG intensity from  $\text{WS}_2$  on an  $\text{Al}_2\text{O}_3$ -Ag film.  $S_0$  is the area of the focused spot of the pumping laser,  $S_{\text{cavity}}$  ( $\sim 75 \times 75 \text{ nm}^2$ ) is the area of the nanocavity. Finally, the average SHG EF is estimated to be 280 in Figure 4a. Since EF should be very sensitive to the spectral overlap, SHG intensity maps in several samples with different Ag nanocubes are measured shown in Figure 4b. The estimated EF is from 42 to 280. This variation of the EF is mainly ascribed to the spectrally shift of SPPR in the nanocavity due to a slight size variation of the Ag nanocubes. The correlation of EF and SPPR under a fixed excitation (shown in Figure S5) also confirms the sensitivity.

Moreover, SHG EF as a function of the fundamental wavelength ( $\lambda_{\text{exc}}$ ) from 790 to 840 nm is measured in the same nanocavity. As shown in Figure 5a (red dots), the maximum



**Figure 5.** (a) Plot of the SHG EF as a function of excitation laser wavelength in the same nanocavity. The red dots and curve represent the experimental data and theoretical calculations, respectively. The normalized linear scattering spectrum of the nanocavity is shown as a gray area. (b) Radiation pattern of the simulated far-field intensity plotted in the  $xz$ -plane. The dipole with a wavelength of 410 nm is used to calculate. The blue area shows the collection cone of the objective.

EF of 220 is obtained at wavelength of 820 nm. Gray area in Figure 5a presents the linear scattering spectrum of the nanocavity with the center wavelength at 410 nm. We found that the largest EF is obtained when the SHG wavelength is matched with the peak of the quadrupole mode of the nanocavity at 410 nm, which is half of the excitation wavelength of 820 nm. When the excitation wavelength moves to 790 nm, EF decreases to  $\sim 50$ . It indicates that the EF is very sensitive to the spectra overlap between linear scattering and SHG spectrum. More similar results are shown in Figure S6.

In order to analyze reasons for the SHG enhancement, the radiation pattern of SHG is simulated, since the angular emission pattern of SHG signals can be modified by the Ag nanocube. The SHG radiation is simulated by placing a dipole

between Ag nanocube and the  $\text{Al}_2\text{O}_3$ -Ag film (see Supporting Information for simulated details). As shown in Figure 5b, red and blue curves present the emission pattern in the  $xz$ -plane with and without Ag nanocube, respectively. Since the collection angular (pale blue area) is determined by an objective with  $\text{NA} = 0.8$ , up to  $\sim 5$ -fold enhancement in SHG collection efficiency is obtained. Note that the charge transfer process between monolayer  $\text{WS}_2$  and Ag nanocubes is prohibited in our experiment, due to the PVP coated the Ag nanocubes. Therefore, the hot carrier injection did not happen which is also useful for SHG enhancement.<sup>46</sup> Additionally, the polarization of the emitted SH signal should also be considered. Because the  $\text{WS}_2/\text{Ag}$  nanostructure shows a quadrupole SH emission while the pure monolayer  $\text{WS}_2$  is a dipole type, which might be influenced by the out-coupling efficiency. We have checked the experiment setup, especially the reflective mirror used for SH emissions. The reflectance of Ag mirror (Thorlabs, PFR10-P01) shows a small variation of 0.3% between p- and s-polarization light at 410 nm. It indicates that the out-coupling efficiency have little influence on the estimated EF.

For further understanding the SHG enhancement mechanism, we develop a theoretical model for the SHG enhancement of nanohybrids based on the theory developed by Singh.<sup>47</sup> The nanohybrid is made of a plasmonic nanocavity doped with quantum emitters (QEs), which is similar to the experimental samples. The monolayer  $\text{WS}_2$  acts as the QE. A strong probe field with frequency  $\omega_1$  is applied to create two-photon absorption in the nanohybrid. The SHG signals are produced by QEs. We consider that the QE is made of two energy levels denoted as  $|1\rangle$  and  $|2\rangle$ , where  $|1\rangle$  is the ground state. The frequency difference between states  $|1\rangle$  and  $|2\rangle$  is denoted as  $\omega_{12}$ . A schematic diagram of the QE is shown in Figure 1a. Surface plasmon polaritons (SPPs) and SPPRs are present at the interfaces of the Ag nanocube and the Ag film. The probe field acts between the transitions  $|1\rangle \leftrightarrow |2\rangle$ , giving us the condition of  $\omega_{12} = 2\omega_1$ . The SPP field of the nanocube and the SPP field produced by the Ag film also act between the transitions  $|1\rangle \leftrightarrow |2\rangle$ . The SHG emission occurs between the transition of  $|1\rangle \leftrightarrow |2\rangle$  due to two photons ( $2\omega_1$ ), giving us the condition  $\omega_{12} = 2\omega_1$ .

We calculate SPP electric field emitted by the Ag nanocube. We denote the dielectric constant of the Ag nanocube as  $\epsilon_c$  and the dielectric constant for the  $\text{WS}_2$  layer (the QE) as  $\epsilon_q$ . We apply a probe field with an amplitude  $E_p$ , frequency  $\omega_1$ , and wavelength  $\lambda$ . When the probe field falls on an Ag nanocube, induced dipole and quadrupole moments are produced in the Ag nanocube. Due to the dipole mode is not resonance with the  $\omega_{12}$  or  $2\omega_1$ , here only the quadrupole mode is calculated. Dipole mode of Ag nanocubes and SPP field produced by the Ag film are calculated in the Supporting Information. The induced quadrupole moment produces the SPP quadrupole electric field. The field can be calculated using the quasi-static approximation.<sup>48–52</sup> Let  $E_{\text{QP}}^{\text{cube}}$  be the electric fields produced by the quadrupole moment. By solving the Maxwell equations in the static wave approximation, the electric field produced by the Ag nanocube is calculated as

$$E_{\text{QP}}^{\text{cube}} = (\Pi_{\text{QP}}^{\text{cube}})E_p, \quad \Pi_{\text{QP}}^{\text{cube}} = \frac{V_{\text{cube}}^{5/3} \epsilon_c^{\text{cube}}}{4\pi r_c^5} \quad (2)$$

where  $r_c$  is the distance between the center of Ag nanocube and the center of the QE, and  $V_{\text{cube}}$  is the volume of the

nanocube, and function  $\zeta_{\text{QP}}^{\text{cube}}$  is called the polarizability factors and depends on the shape of the Ag nanocube.

Following the method of ref 47, the dipole moment in the QEs,  $P_{\text{qd}}$ , due to two-photon absorption has been calculated in reference<sup>53,54</sup> and is written as

$$P_{\text{qd}} = k_{12}\rho_{12}(1 + \Pi_{\text{QP}}^{\text{cube}})E_{\text{p}}e^{-i2\omega_1 t} = k_{12}\rho_{12}L(2\omega_1)E_{\text{p}}e^{-i2\omega_1 t} \quad (3)$$

where  $k_{12}$  is the two-photon coupling constant due to the transition and  $\rho_{12}$  is the two-photon density matrix element.  $L(2\omega_1)$  is defined as the enhanced electric field caused by Ag nanocavity. Neglecting the dipole-SPP field in the Ag nanocube and SPP field produced by the Ag film, we get the SHG electric field produced by the QE as

$$E_{\text{SHG}}^{\text{QE}} = \frac{k_{12}\rho_{12}L(2\omega_1)E_{\text{p}}e^{-i2\omega_1 t}}{(4\pi r_{\text{qd}}^3/3)\epsilon_{\text{b}}} \quad (4)$$

We calculated  $\rho_{12}$  due to two-photon emission by using the density matrix method.<sup>47,55</sup> We found the following expression for  $\rho_{12}$  as

$$\rho_{21} = \frac{i\Omega_{21}L(2\omega_1)}{(\gamma_2/2 - i\delta_{21} - i\Delta_s)} \quad (5)$$

The parameter  $\delta_{12} = 2\omega_1 - \omega_{12}$  is called the probe detuning factor. The parameter  $\Omega_{21}$  is called the Rabi frequencies. The parameter  $\gamma_2$  is the spontaneous decay rate for state  $|2\rangle$ . The parameter  $\Delta_s$  is the Stark frequency and defined in ref 47. The intensity of the SHG electric field generated by the QE in the cavity is calculated as

$$I_{\text{SHG}}^{\text{cavity}} = (1/2)\epsilon_0 c \sqrt{\epsilon_{\text{b}}} |E_{\text{SHG}}^{\text{QE}}|^2 = \frac{(1/2)\epsilon_0 c I_{\text{p}} \sqrt{\epsilon_{\text{b}}}}{(4\pi r_{\text{qd}}^3/3)^2 \epsilon_{\text{b}}^2} \left| \left( \frac{ik_{12}\Omega_{21}L^2(2\omega_1)}{(\gamma_2/2 - i\delta_{21} - i\Delta_s)} \right)^2 \right| \quad (6)$$

where  $I_{\text{p}}$  is the intensity of probe field. Similarly, we can calculate the SHG intensity when metallic nanocubes are absent. In other words, the hybrid structure is made of a metallic film with a spacer layer and monolayer WS<sub>2</sub> is placed on the spacer layer. In this case SHG intensity is denoted by  $I_{\text{SHG}}^{\text{film}}$ . The SHG intensity for this hybrid structure can be calculated from eq 6 by putting  $\Pi_{\text{QP}}^{\text{cube}} = 0$ . Let us calculate the SHG enhancement ratio due to presence of metallic nanocubes. It is defined as the ratio of  $I_{\text{SHG}}^{\text{cavity}}$  and  $I_{\text{SHG}}^{\text{film}}$  as  $I_{\text{ratio}}^{\text{SHG}} = I_{\text{SHG}}^{\text{cavity}}/I_{\text{SHG}}^{\text{film}}$ . Finally, the expression for the SHG ratio can be obtained as  $I_{\text{ratio}}^{\text{SHG}} = L^4(2\omega_1)$ , indicating that the SHG EF is proportional to the square of the local-field intensity in Ag nanocavity at SHG wavelength.

We have performed numerical simulations on the intensity of the SHG field produced by the nanohybrid. In simulations we have taken the dielectric constant of WS<sub>2</sub> as  $\epsilon_{\text{q}} = 6.2$ . The length of each side of the Ag-nanocube is about 75 nm. The plasmon frequency and the relaxation time for Ag are  $\omega_{\text{p}} = 8.9$  eV and  $\tau = 1.60 \times 10^{-14}$  s.<sup>56</sup> The value of the dielectric constant is  $\epsilon_{\infty} = 5$ .<sup>56</sup> Using these physical parameters we have calculated SPP resonance wavelengths of Ag nanocube (shown in Figure S7) and found as  $\lambda_{\text{DIP}}^{\text{cube}} = 620$  nm and  $\lambda_{\text{QP}}^{\text{cube}} = 410$  nm, respectively. These values agree well with experimental values. The SPP resonance wavelength due to the metallic film is also calculated and found as  $\lambda_{\text{sp}}^{\text{film}} = 120$  nm (see the Supporting Information for more calculated details).

In Figure 5a, the calculated enhancement ratio ( $I_{\text{ratio}}^{\text{SHG}}$ ) as a function of  $\lambda_{\text{exc}}$  is plotted by a red curve. It shows that a good

agreement between the theory and the experimental data. Note that the data in theoretical and experimental is a qualitative basis but not a quantitative basis. In general, under harmonic resonance in pure metal NPs, enhanced SHG is attributed to the improved emission efficiency, called antenna effect. The SHG intensity is in proportional to the local-field intensity, written as  $I_{\text{ratio}}^{\text{SHG}} \propto L^2(2\omega_1)$ .<sup>28</sup> Similar to the antenna effect, the SHG emission efficiency from a monolayer WS<sub>2</sub> is improved by the enhanced local-field  $L(2\omega_1)$ , as shown in eq 4 in the monolayer WS<sub>2</sub>-Ag nanocavity nanohybrids. More importantly, unlike pure metal arrays, the source of SHG ( $P_{\text{SHG}}$ ) in nanohybrids is also enhanced, which is shown in the enhanced density matrix element in eq 5. It is also proportional to the local-field intensity. Therefore, under two cooperative effects, harmonic resonance enhanced SHG intensity in nanohybrid is in proportion to the square of the local-field intensity. In experiment, a 2-order of magnitude enhancement can be achieved. In comparison with the harmonic resonance enhanced SHG in pure metal NPs, a novel SHG enhancement mechanism is presented. It is known that SHG signals actually contain not only dipolar terms but also multipolar ones, such as quadrupolar response. However, the relative magnitude of contributions from the quadrupole with respect to the dipole is generally quite weak. Therefore, our finding also can be used for enhancing the quadrupole signal by dipole-quadrupole interaction in hybrid nanostructure, which has potential implications for directional nonlinear generation.<sup>57</sup>

In conclusion, the harmonic resonance enhanced SHG from a monolayer WS<sub>2</sub> coupled to a plasmonic nanocavity is studied experimentally and theoretically. The nanocavity consists of single Ag nanocubes and an Al<sub>2</sub>O<sub>3</sub>-Ag film. A monolayer WS<sub>2</sub> is placed between the Ag nanocubes and the Al<sub>2</sub>O<sub>3</sub> film. When the resonance wavelength of nanocavity is overlapped well with the SHG wavelength at 410 nm from the monolayer WS<sub>2</sub>, a  $\sim 300$ -fold SHG enhancement is achieved. For understanding the SHG enhancement mechanism, the quantum mechanical density matrix method has been used to develop a theory for the SHG. It is found that SHG intensity is proportional to the square of the local-field intensity of Ag NPoM nanocavity under the dipole-quadrupole interaction. Specifically, the source of SHG in the monolayer WS<sub>2</sub> is enhanced in proportional to the local-field intensity of Ag nanocavity. While the SHG emission efficiency can also be enhanced by the local-field intensity, known as antenna effect. It presents a novel SHG enhancement mechanism, which has potential applications in nonlinear devices and hybrid nonlinear structures.

## ■ ASSOCIATED CONTENT

### Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acsp Photonics.9b01499>.

The Raman spectrum of the WS<sub>2</sub> flake; The size distribution of the Ag nanocubes; Simulations of scattering, Scattering spectra of nanocavity before and after SHG measurement; Correlation between SHG EF and the scattering peak; Polarization property and SH collection efficiency of single nanocavities; The details of theoretical modeling; Estimation of the SHG conversion efficiency (PDF)

## AUTHOR INFORMATION

### Corresponding Authors

**Kai Wang** – Wuhan National Laboratory for Optoelectronics and School of Physics, Huazhong University of Science and Technology, Wuhan 430074, China; [orcid.org/0000-0003-2122-1294](https://orcid.org/0000-0003-2122-1294); Email: [kale\\_wong@hust.edu.cn](mailto:kale_wong@hust.edu.cn)

**Mahi R. Singh** – Department of Physics and Astronomy, The University of Western Ontario, London N6G 3K7, Canada; [orcid.org/0000-0003-0930-4003](https://orcid.org/0000-0003-0930-4003); Email: [msingh@uwo.ca](mailto:msingh@uwo.ca)

**Peixiang Lu** – Hubei Key Laboratory of Optical Information and Pattern Recognition, Wuhan Institute of Technology, Wuhan 430205, China; Wuhan National Laboratory for Optoelectronics and School of Physics, Huazhong University of Science and Technology, Wuhan 430074, China; CAS Center for Excellence in Ultra-intense Laser Science, Shanghai 201800, China; Email: [lupeixiang@hust.edu.cn](mailto:lupeixiang@hust.edu.cn)

### Authors

**Xiaobo Han** – Hubei Key Laboratory of Optical Information and Pattern Recognition, Wuhan Institute of Technology, Wuhan 430205, China

**Patrick D. Persaud** – Department of Physics and Astronomy, The University of Western Ontario, London N6G 3K7, Canada

**Xiangyuan Xing** – Wuhan National Laboratory for Optoelectronics and School of Physics, Huazhong University of Science and Technology, Wuhan 430074, China

**Weiwei Liu** – Wuhan National Laboratory for Optoelectronics and School of Physics, Huazhong University of Science and Technology, Wuhan 430074, China; [orcid.org/0000-0001-9451-1968](https://orcid.org/0000-0001-9451-1968)

**Hua Long** – Wuhan National Laboratory for Optoelectronics and School of Physics, Huazhong University of Science and Technology, Wuhan 430074, China

**Fang Li** – Hubei Key Laboratory of Optical Information and Pattern Recognition, Wuhan Institute of Technology, Wuhan 430205, China

**Bing Wang** – Wuhan National Laboratory for Optoelectronics and School of Physics, Huazhong University of Science and Technology, Wuhan 430074, China

Complete contact information is available at:

<https://pubs.acs.org/10.1021/acsp Photonics.9b01499>

### Notes

The authors declare no competing financial interest.

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